Air- and water-stable and photocatalytically active germanium-based 2D perovskites by organic spacer engineering

Metal halide perovskite photocatalysis is widely studied in several applied fields. In this work, Romani et al. demonstrate 2D Ge-containing perovskite engineering to improve air and water resistance. Such characteristics are then exploited in solar-driven hydrogen generation in an aqueous environment.
**SUMMARY**

There is increasing interest in the role of metal halide perovskites for heterogeneous catalysis. Here, we report a Ge-based 2D perovskite material that shows intrinsic water stability realized through organic cation engineering. Incorporating 4-phenylbenzilammonium (PhBz) we demonstrate, by means of extended experimental and computational results, that PhBz$_2$GeBr$_4$ and PhBz$_2$GeI$_4$ can achieve relevant air and water stability. The creation of composites embedding graphitic carbon nitride (g-C$_3$N$_4$) allows for a concept for light-induced hydrogen evolution in an aqueous environment by 2D Ge-based perovskites thanks to the effective charge transfer at the heterojunction between the two semiconductors.

**INTRODUCTION**

Metal halide perovskites (MHPs) are attracting huge interest for their possible application in heterogeneous photocatalysis following the recent synthesis of materials and heterostructures that have been found to be efficient for a plethora of photocatalyzed chemical reactions, including hydrogen generation and CO$_2$ reduction.\(^1\)\(^{–}\)\(^17\)

A major limitation of MHPs in photocatalysis is their limited water stability, which derives from the high ionic character of the metal halide framework.\(^2\)\(^,\)\(^18\)\(^,\)\(^19\) In this context, 2D-layered MHPs may be particularly suitable to overcome this issue in virtue of the vast range of organic spacers that can be inserted as a protective barrier between the inorganic perovskite layer(s).\(^2\)\(^,\)\(^20\) In fact, upon the introduction of a highly hydrophobic organic spacer in the perovskite structure, significant moisture and water stability could be achieved, and examples exist of systems forming a suspension in water instead of being dissolved.\(^8\)\(^,\)\(^21\)\(^–\)\(^23\) In this respect, recently, some Bi- and Sn-based perovskites with improved water resistance were found to show significant visible-light photocatalytic activity of hydrogen photogeneration and organic dye degradation, which was further enhanced by designing heterostructures with graphitic carbon nitride (g-C$_3$N$_4$).\(^8\)\(^,\)\(^2\)\(^,\)\(^24\)\(^,\)\(^25\) Similar strategies have been used also by other authors on the Cs$_3$Bi$_2$I$_9$ perovskite derivative.\(^7\)\(^,\)\(^11\)

While materials engineering has been successful for the synthesis of lead-free MHP photocatalysts, there are no reports about any photocative Ge halide perovskite. Recently, a series of Ruddlesden-Popper (RP) 2D Ge bromide perovskites (n = 1), namely A$_2$GeBr$_4$, with A = C$_6$H$_4$CH$_2$CH$_2$NH$_3$ (phenylethylammonium [PEA]); BrC$_6$H$_4$CH$_2$CH$_2$
NH$_3$ (Br-phenylethylammonium [BrPEA]); FC$_6$H$_4$CH$_2$CH$_2$NH$_3$ (F-phenylethylammonium [FPEA]); and C$_6$H$_4$CH$_2$NH$_3$ (benzylammonium [BzA]), showed air stability but not water tolerance. Therefore, we further extended this quest by designing a 2D composition including the 4-phenylbenzilammonium (PhBz) spacer cation (Figure 1A). This cation presents extended π-conjugated systems, creating a unique condition in which strong intra-layer van der Waals (vdW) interactions are established, substantially stabilizing the resulting bulk perovskite. Thanks to this stabilization, we demonstrate improved air and water stability in 2D Ge perovskites containing the PhBz cation. Such stability is exploited in the construction of heterojunctions with g-C$_3$N$_4$ and their application in the solar-driven hydrogen production in aqueous environment.

RESULTS AND DISCUSSION
PhBz$_2$GeBr$_4$ perovskite structure and stability
The novel PhBz$_2$GeBr$_4$ perovskite was synthesized by means of wet-chemistry route as described in the experimental procedures. The powder has a white color (Figure 1B), and the corresponding high-resolution synchrotron radiation (SR) powder X-ray diffraction (XRD) pattern collected at 0.3547 Å is shown in Figure 1C. The indexing of the SR-XRD data (Figure S1A) provided an orthorhombic cell (space group $Pmc2_1$) with refined lattice parameters of $a = 6.0260(1)$, $b = 9.1744(2)$, and
c = 44.376(1) Å. There are no literature reports on 2D perovskites containing PhBz cation, therefore a direct comparison with analogous compositions is not possible. The most similar material reported, in terms of organic cation, is BPEA2PbI4 (BPEA = 2-(4-biphenyl)ethylamine), in which the organic cation has an ethyl linked to the amine group.25 Even though the central metal and the halide are different with respect to PhBz2GeBr4, the agreement with the symmetry of the BPEA2PbI4 crystal suggests a similar organic spacer arrangement as shown in Figure S1B, where PhBz2GeBr4 diffraction is plotted against the expected Bragg peaks from BPEA2PbI4. The long c axis is, as well, in a similar range as the one reported here.25

The optical properties of PhBz2GeBr4 have been determined by UV-visible (UV-vis) absorption spectroscopy and photoluminescence (PL) (see Figures 1D and 1E). From the Tauc plot, we estimate a band gap of 3.64 eV, while the PL spectrum shows a structured intense band composed of three contributions with the main emission peak centered at about 415 nm.

To test the stability versus moisture of the synthesized material, as-prepared PhBz2 GeBr4 (kept under argon in glovebox) was exposed to laboratory air (T = 22°C, relative humidity [RH] = 35%) for 2 months. The XRD pattern collected after this time interval shows a perfect match with the pattern of the as-prepared PhBz2GeBr4 (cf. red and black lines in Figure 2A). Furthermore, PhBz2GeBr4 powders were dispersed in water and kept under stirring for 24 h, and the diffraction pattern of the powders was found in very good agreement with the as-prepared material (cf. blue versus black patterns in Figure 2A). In the rescaled inset of Figure 2A, we highlighted the peaks around 10°–40° to put in prominence the very good agreement between the patterns in the whole diffraction range.

Further evidence of water stability has been assessed through a leaching test. In this assay, PhBz2GeBr4 has been placed in water under magnetic stirring, and the amount of Ge in the solution has been determined by inductively coupled plasma
optical emission spectroscopy (ICP-OES) (see details in the supplemental information). After 4, 8, 16, and 24 h of stirring, the Ge amounts detected in the solution were 4.3%, 5.7%, 5.9%, and 6%, respectively, of the total amount of Ge present in the perovskites, thus confirming the high water stability and insolubility of PhBz₂GeBr₄.

In addition to the clear evidence of air and water structural stability reported above, we applied X-ray photoelectron spectroscopy (XPS) on the as-prepared sample and on the sample after immersion in water. The whole set of recorded spectra is shown in Figure S2, while in Figure 2B, we report the Ge 3d spectrum. The Ge 3d spectra of the two samples reported in Figure 2B are superimposable (as for all the other elements probed, cf. supplemental information), indicating that the Ge ions in immersed samples retain the same oxidation state of as-prepared materials.

The relative position of the Ge 3d peak suggests the possible presence of Ge(IV) on the surface of the material, possibly in the form of a native oxide, which could act as protective layer as recently proposed to justify an enhanced air stability of 2D and 3D Ge perovskites. More specifically, the binding energy position of the Ge 3d peaks at 33.4 eV is indicative of a Ge(IV) state and an additional indication of the Ge(IV) oxide formation at the surface of the material derives from the curve-fit analysis of the corresponding O1s spectra (Figure S2), revealing the presence of a component at 531.8 eV typical of GeO₂.

However, since previously synthesized Ge-based 2D perovskites did not show any water stability, the reason of the improved water stability of PhBz₂GeBr₄ may predominantly lay in the bulk properties originating by the significant steric hindrance and hydrophobicity of the organic spacer. Similar water stability has been also observed in some Sn-based MHPs, namely DMASnBr₃ and PEA₂SnBr₄. In the first material, the mechanism leading to improved stability and effective photocatalytic activity results from the valence band energy stabilization, which should reduce the exposure of the material to oxidants, restoring a value similar to MAPbI₃, healing the material from degradation into Sn⁴⁺ phases and self-p-doping effects. On the other hand, the improved stability in PEA₂SnBr₄ was again related to the nature of surface-terminating groups and the presence of hydrophobic PEA moiety. For lead-based perovskites, the highly ionic nature of 3D systems has hampered their use in aqueous environments. For MAPbI₃, for example, any photocatalytic activity has been only reported in hydrohalic acids by exploiting the dynamic equilibrium of the dissolution and precipitation of the perovskite in saturated aqueous solutions. More water-stable Pb-containing perovskites have been obtained by exploring the use on long aliphatic chains in 2D systems such as for (HDA)₂PbI₄ (HAD = hexadecylammonium). This phase was also used in photocatalytic activity for photoredox C–C bond cleavage and dehydrogenation catalysis but not for hydrogen photogeneration.

PhBz₂GeBr₄ hydrogen generation experiments

We exploited the advantage of PhBz₂GeBr₄ superior water stability by testing its possible application in photocatalysis. Hydrogen photogeneration characteristics have been determined according to a commonly employed protocol we already applied for other MHPs. First, we determined the hydrogen evolution rate (HER) of pure PhBz₂GeBr₄ under simulated solar light, which turned out to be ~6 μmol g⁻¹ h⁻¹, a low but relevant value for a pure MHPs, also considering the relatively high band gap of the materials, 3.6 eV, which corresponds to efficient...
absorption only in the UV region of the light spectrum (in line with the UV-vis absorption spectrum; Figure 1C). To enhance the photoactivity of the perovskite, we prepared composites with a well-known visible-light-absorbing semiconductor, namely g-C₃N₄. Composites have been synthesized by means of wet-chemistry route as reported in the experimental procedures at different weight percentages (% wt) of MHP relative to g-C₃N₄ (1%, 2.5%, 3.5%, 5%, and 15%). Morphologies of pure samples and composites have been assessed by scanning electron microscopy (SEM) and are reported in Figure S3. Figure 3A reports the XRD patterns of the composites showing the main contribution of g-C₃N₄ up to 15 wt %. We point out that the main peak of PhBz₂GeBr₄, located around 4°, is already detectable in the sample containing 2.5 wt % of perovskite. UV-vis spectra of the composites (Figure 3B) show a significant contribution to absorbance deriving from the carbon nitride.

Figure 3. Structural and optical properties of PhBz₂GeBr₄/g-C₃N₄ composites
(A) XRD patterns of PhBz₂GeBr₄/g-C₃N₄ composites for different perovskite loadings.
(B) UV-vis spectra of PhBz₂GeBr₄/g-C₃N₄ composites for different perovskite loadings.
(C) Normalized emission spectra of PhBz₂GeBr₄/g-C₃N₄ composites at different percentages of perovskite loading (wt %). g-C₃N₄ refers to pristine material. λₑₓc = 320 nm.
(D) Normalized PL decays for the same composites (λₑₓc = 320 nm; λₑₓm = 450 nm). In the inset, PLQY for all samples obtained with continuous wave (CW) excitation at 405nm.
The normalized PL spectra of pure compounds and composites are reported in Figure 3C. Both pristine materials show intense luminescence bands, with PhBz₂GeBr₄ peaking at 415 nm and g-C₃N₄ at 455 nm. Their bands falling in different spectral regions allow us to monitor the contribution of the single components to the PL features of the composites. Noticeably, the composites, when excited at a wavelength that allows the absorption of both materials, show a spread PL emission overshadowed by g-C₃N₄ contribution. g-C₃N₄ emission is energetically spread spanning from 410 to 560 nm due to various deactivation paths existing within the energy band diagram of the material. The PL quantum yields (PLQYs) as well as the lifetimes (τ) (Figure 3D) are very similar for pristine and for g-C₃N₄-containing composites. The PLQY varies from 3.9% ± 0.4% to 5.2% ± 0.5%, and the average τ is 9.0 ± 0.5 ns for all specimens, further suggesting the central role played by carbon nitride states in the excitation deactivation path, while the PLQY of pure PhBz₂GeBr₄ is lower (2.2% ± 0.2%), compatible with the shorter PL lifetime. These observations nicely fit the band structure of the junction, which is modeled below (vide infra).

The normalized PL spectra show, however, some differences among the samples. Features of g-C₃N₄ dominate the spectra at high perovskite loadings, suggesting a highly efficient energy transfer from the perovskite moiety to g-C₃N₄. The low-loading composites show instead some minority features attributable to residual perovskite contribution (Figure 3C). For these compositions, namely 1%, 2.5%, and 3.5% loading of perovskite, the energy transfer appears somehow less effective, suggesting a possible defect-filling mechanism at the g-C₃N₄/perovskite interface (see later in the text). An additional difference related to the shape of the composite emission can be found in the region of its peak (around 440–460 nm); here, the g-C₃N₄ and the high-loading composites show a more intense contribution at low energies, peaking their emission at 458 nm; meanwhile, low-loading composites (1% and 2.5%) show the maximum emission at 444 nm. The shape of the emission band in g-C₃N₄ materials can be attributed to the relative population of diverse energy transitions, withstanding a complex deactivation path for the excitation in g-C₃N₄ materials; in particular, the one at low energies are connected to π*-π transition, while the one at high energies are related to deactivation of the nitrogen atom lone pair. In our systems, the diverse shapes suggest how differences in the relative populations of excitation/deactivation processes exist between low- and high-loading composites. The peculiar optical behavior of composites appears to be a distinctive trait of perovskite carbon nitride systems, as already described elsewhere, suggesting that low levels of perovskite doping may act on funneling the charges upon localized states. Those active sites could be fundamental in boosting the photocatalytic performances of low-loading composites. Such effect faded with increased perovskite loadings.

Finally, we investigated the solar-driven catalytic efficiency of the prepared composites in terms of the HER. The measurements have been carried out by employing current protocols applied in the literature for g-C₃N₄-based composites, i.e., in 10% (v/v) aqueous triethanolamine (TEOA), as a typical sacrificial agent and with Pt (3 wt %) as metal co-catalyst. Figure 4A shows the HER as a function of perovskite loading, while Figure 4B shows the kinetics of the hydrogen evolution for the best-performing composite from Figure 4A, namely 2.5 wt % PhBz₂GeBr₄/g-C₃N₄.

Noticeably, the composites display a synergic effect, providing a significant improvement of the HER of about 8 times with respect to pure carbon nitride (81 μmol g⁻¹ h⁻¹) and of about 100 times with respect to pure PhBz₂GeBr₄ (6 μmol g⁻¹ h⁻¹). The maximum HER is found at a relatively low MHP loading of 2.5 wt %, while higher perovskite percentages are not beneficial to improve the reaction. For the optimal
composite, namely at 2.5 wt %, the kinetics of H₂ evolution, reported in Figure 4B, indicates a substantial linear increase of the hydrogen production as a function of time. The value of apparent quantum yield (AQY%), calculated as the percent ratio H₂ moles/incident photons moles, for this composition (2.5 wt % of perovskite) was 5.2%. This is the first evidence of the application of a Ge-based perovskite in solar-driven hydrogen generation; therefore, any comparison with pre-existing literature is not possible. However, the measured rates are similar to those measured for Sn- and Bi-based perovskites. The composite at 2.5 wt % of PhBz₂GeBr₄ has been tested over four successive catalytic cycles by centrifugating, recovering, and subjecting the sample to the same photocatalytic procedure. The HER in the second cycle was about 94% of the initial HER, while in both the third and fourth cycles, it was reduced to about 74% (Figure S4); such a result could be related to the slight, but detectable, Ge leaching reported above. Finally, the catalyst was recovered after a photogeneration test and analyzed by XRD to test the material stability. For the sake of clarity this test was accomplished on the material with 15 wt % loading of PhBz₂GeBr₄ since it showed the most evident reflections from the perovskite. The patterns of fresh PhBz₂GeBr₄ 15 wt %/g-C₃N₄ and of the same sample recovered after 6 h of irradiation under the conditions reported above are shown in Figure S5, indicating a very good stability of the composite after the photocatalytic test. To get a further insight into the microscopic mechanism underlying the HER behavior, we performed transient absorption spectroscopy (TAS) measurements on the composites.

Differential transmission (DT/T) measurements were performed on thin films of different PhBz₂GeBr₄/g-C₃N₄ composites on quartz substrate, which we exploit to probe carrier dynamics. Note that, oftentimes, the technique is referred to as TAS since DT/T and absorption are linked and can be converted into each other. DT/T allows us to detect the changes in optical absorption that are induced by ultrafast laser pulses, in our case constituted by 100 fs-long pulses that are 320 nm in
wavelength. DT/T is then determined with a pump-and-probe technique where fs supercontinuum white pulses are delayed with respect to pump pulses with an adjustable delay line (see the experimental section in the supplemental information). From the results collected in Figure 5, two broad DT/T features are visible in g-C$_3$N$_4$ sample (gray line): a positive bleaching band around 500 nm, resulting from filling of the excited state, and a negative band, which peaked around 650 nm, that can be associated with photoinduced absorption from the excited state. When adding PhBz$_2$GeBr$_4$ from 1 to 3.5 wt %, the 500 nm-bleaching band is gradually replaced by a lower-energy one centered around 650 nm. In contrast, for % wt >5, a photoinduced band, similar to pure g-C$_3$N$_4$, is partially recovered.

The simultaneous fading of photoinduced signal/high-energy bleaching and the appearance of low-energy bleaching in DT/T spectra may be interpreted as evidence for defect filling at the g-C$_3$N$_4$/perovskite interface, possibly due to charge transfer. Both PL band structure and bleaching in DT/T can be therefore correlated with partial charge transfer to g-C$_3$N$_4$, which happens to be more efficient for intermediate compositions (1%–3.5% loading), leading to better photocatalytic performance. At variance with this, samples with above 5 wt % have PL and DT/T spectra more similar to g-C$_3$N$_4$, an indication that most photoexcited electrons are kept in g-C$_3$N$_4$ states, leading to either radiative recombination or to photoinduced absorption. Such a trend with respect to perovskite loading is compatible with what has been already observed in a Bi-based perovskite composite for photocatalysis. The trend of both a DT/T bleaching feature and hydrogen production rate can be linked with defect filling at the g-C$_3$N$_4$/perovskite interfaces but with a disclaimer: the defect states involved in absorption process, affecting DT/T dynamics, are substantially not taking part in PL emission, which results instead from g-C$_3$N$_4$ states. In fact, the results of time-resolved PL measurements show that the PL lifetime of g-C$_3$N$_4$/perovskite compounds is not different from that of g-C$_3$N$_4$ and is not varying with perovskite loading, despite pure perovskite showing a much shorter lifetime than g-C$_3$N$_4$ compounds (see Figure 3D).

**Computational modeling**

The origin of the reported results was further investigated from an atomistic perspective with the aim of understanding (1) the outstanding water stability of...
the Ge-based perovskites synthesized in this work, and (2) the efficient photocatalytic production of \( \text{H}_2 \) when the perovskites are used in a composite with g-C\(_3\)N\(_4\) (see Note S1). In addition, computational modeling was extended to also include the iodide-analogous phase, namely PhBz\(_2\)GeI\(_4\), to determine its possible application in photocatalysis. As a matter of fact, the presence of iodide can red shift the band gap and provide a more effective charge transfer in the heterostructure.\(^{20,24,26}\) However, iodide-based materials are known to be less stable than the bromide counterparts.

To fulfill the first goal, we calculated the bulk formation energies \( E_f(\text{bulk}) \) of the water-stable Ge-based perovskites, and we compared them with those of previously synthetized materials, bearing different A-site spacer cations, which were found to dissolve in aqueous environment.\(^{24}\) Since no experimental crystallographic structure is currently available for PhBz\(_2\)GeI\(_4\) and PhBz\(_2\)GeBr\(_4\), we consider the perovskites with BPEA, which differs from PhBz only for an extra CH\(_2\) between the aromatic ring and the ammonium moiety.\(^{25}\) BPEA\(_2\)GeI\(_4\) and BPEA\(_2\)GeBr\(_4\) were modeled starting from the analogous BPEA\(_2\)PbI\(_4\) as described in the supplemental information. We consider, for our comparison, PEA\(_2\)GeI\(_4\) and BrPEA\(_2\)GeBr\(_4\), which have been synthetized and characterized in Chiara et al.\(^{24}\) The details of the structural models are reported in the supplemental information.

From the following reaction

\[
2AX + \text{GeX}_2 \rightarrow A_2\text{GeX}_4,
\]  

(Equation 1)

where A = PEA, BrPEA, and BPEA and X = Br and I, we define \( E_f(\text{bulk}) \) as follows:

\[
E_f(\text{bulk}) = E(A_2\text{GeX}_4) - E(AX) - E(\text{GeX}_2).
\]

(Equation 2)

In Equation 2, \( E(A_2\text{GeX}_4) \), \( E(AX) \), and \( E(\text{GeX}_2) \) are the total energies of \( A_2\text{GeX}_4 \), AX, and GeX\(_2\), which are calculated from their respective atomistic models (cf. Note S1). Results collected in Table 1 clearly indicate that perovskites bearing BPEA as spacer A-site cation are substantially more stable than the others, with formation energies being up to 0.7 eV lower. Such a larger stability may be ascribed to the enhanced vDW interactions available within the A-cation layers separating the inorganic frameworks when using the larger BPEA molecule. In order to further verify this consideration, we calculate for each studied material the formation energy of a neutral A vacancy, \( V_A \), which is defined as

\[
E_f(V_A) = E(V_A) - E(A_2\text{GeX}_4) - E(A),
\]

(Equation 3)

where \( E(V_A) \) is the total energy of the perovskite model with a missing A molecule and \( E(A) \) the total energy of an isolated A molecule. From Table 1, we evince that the energy associated with the removal of an A cation from the bulk perovskite is remarkably higher (up to 1 eV) for BPEA when compared with the other systems, thus clearly indicating that intra-layer vDW interactions sizably stabilize BPEA perovskites. Furthermore, the higher solvation Gibbs free energies calculated for BPEA and PhBz (cf. Table S4) indicate a reduced tendency of these cations to dissolve in water when

<table>
<thead>
<tr>
<th>Perovskite</th>
<th>( E_f(\text{bulk}) )</th>
<th>( E_f(V_A) )</th>
<th>Water stable?</th>
</tr>
</thead>
<tbody>
<tr>
<td>PEA(_2)GeBr(_4)</td>
<td>-4.15</td>
<td>3.98</td>
<td>no</td>
</tr>
<tr>
<td>BrPEA(_2)GeBr(_4)</td>
<td>-4.12</td>
<td>4.18</td>
<td>no</td>
</tr>
<tr>
<td>(BPEA)(_2)GeI(_4)</td>
<td>-4.65</td>
<td>4.92</td>
<td>yes</td>
</tr>
<tr>
<td>(BPEA)(_2)GeBr(_4)</td>
<td>-4.80</td>
<td>5.26</td>
<td>yes</td>
</tr>
</tbody>
</table>
compared with the cations employed in previous work.\textsuperscript{24} We note that such a result is in line with a previous observation of the increased water stability observed when replacing methylammonium with dimethylammonium in tin perovskites.\textsuperscript{31,37}

To study the beneficial effect on the photocatalytic \( \text{H}_2 \) production of the (BPEA)\(_2\) GeX\(_4\) perovskites in a composite with g-C\(_3\)N\(_4\), we investigated their band alignment. To this end, we constructed atomistic models of the surface of these materials. When considering the (100) surface of (BPEA)\(_2\)GeX\(_4\), we find that the most stable termination is the stoichiometric (BPEA)\(_X\)-terminated one, with the BPEA cations pointing their hydrophobic biphenyl moieties toward the surface (cf. Figure 6A for the (BPEA)\(_I\)-terminated surface of (BPEA)\(_2\)GeI\(_4\)), while the ammonium moieties interact with the subsurface inorganic chain. This termination features a surface energy as small as 0.005 eV/\( \text{Å}^2 \) (cf. supplemental information for details of the calculations).

Then, we employed advanced electronic-structure calculations to evaluate the band gap of the materials and align the band edges with respect to the vacuum level (cf. Note S1 and Tables S2 and S3). The band alignment for (BPEA)\(_2\)GeI\(_4\) and (BPEA)\(_2\)GeBr\(_4\) is reported in Figure 6B along with that previously calculated in Romani et al.\textsuperscript{9} for g-C\(_3\)N\(_4\). The band edges of (BPEA)\(_2\)GeI\(_4\) are found be favorably aligned with respect to those of g-C\(_3\)N\(_4\), thus promoting the transfer of photogenerated charges in a type 2 heterojunction, which could also improve the carrier lifetime. At variance with this, (BPEA)\(_2\)GeBr\(_4\), which features a larger band gap (3.7 versus 2.81 eV), has a valence band edge at an energy 0.15 eV lower than g-C\(_3\)N\(_4\), providing a type I heterojunction. This implies that hole transfer from g-C\(_3\)N\(_4\) to the perovskites might be subject to a small energy barrier, which may reduce the efficiency of the composite.

The results of the computational modeling have put in prominence two main results: (1) confirmation of the water stability for BPEA\(_2\)GeBr\(_4\) together with a beneficial band
alignment with g-C₃N₄, as demonstrated by the above reported hydrogen photogeneration experiments, and (2) prediction of good water stability by BPEA₂GeI₄ (even though it has lower $E_f^{(bulk)}$ and $E_f^{(VA)}$ with respect to BPEA₂GeBr₄) and possible superior photocatalytic performance due to a better band alignment with respect to the bromide-containing counterpart.

**PhBz₂GeI₄ hydrogen generation experiments**

To test this last computational evidence, we synthesized the PhBz₂GeI₄ perovskite and a series of composites with g-C₃N₄ (1%, 2.5%, 3%, 3.5%, 4%, 4.5%, 5%, and 15% of perovskite loading). Figures S6 and S7 report the XRD and absorbance data on the composite. We used the same approach we employed for PhBz₂GeBr₄ in testing the air and water stability of PhBz₂GeI₄. Figure S8 also shows good air and water stabilities for the iodide-containing perovskites according to the computational prediction. However, the leaching test revealed an amount of Ge in the solution of about 40% already after 4 h of stirring. This value remains constant after 8 and 16 h of stirring (42% and 43%), suggesting the possible occurrence of a solubility equilibrium that then keeps the perovskite unchanged in the solution according to the XRD diffraction after recovering the powder (Figure S8). Substantially the same leaching (44%) was noticed also under photocatalytic conditions, turning in line with the overall good photocatalytic performance of this composite, described hereafter. Even though the PhBz₂GeI₄ is less stable in water, which also agrees with the calculated values of $E_f^{(bulk)}$, $E_f^{(VA)}$, we performed a thorough investigation of the hydrogen photogeneration efficiency. Figure 7A reports the HER as a function of perovskite loading for the PhBz₂GeI₄/g-C₃N₄ composites, while Figure 7B shows the kinetics of the hydrogen photogeneration for the best-performing composite from Figure 7A, namely 3 wt % PhBz₂GeI₄/g-C₃N₄. For this system, the AQY% is 2.8%, which is lower with respect to the best-performing PhBz₂GeBr₄/g-C₃N₄ composite. The higher HER with a lower AQY (calculated as the percent ratio $H_2$ moles/incident photons moles) is in line with the presence of iodide extending the absorption in the visible part of the spectrum, allowing it to harness more photons. The overall result for the HER of PhBz₂
GeBr₄/g-C₃N₄ composites is observed to be higher, but the AQY is not, for a diverse absorptivity of the specimens.

The HER for the PhBz₂GeI₄/g-C₃N₄ composites is higher with respect to the PhBz₂ GeBr₄/g-C₃N₄ composites, reaching a value of about 1,200 μmol g⁻¹ h⁻¹, again with a synergic effect with respect to pure carbon nitride (81 μmol g⁻¹ h⁻¹), and of about 600 times with respect to pure PhBz₂GeI₄ (2 μmol g⁻¹ h⁻¹). The maximum HER is found at 3 wt %, very close to the value of 2.5 wt % found for the PhBz₂ GeBr₄/g-C₃N₄ series. Again, according to the computational modeling, we could confirm the better band alignment of this perovskite with g-C₃N₄, leading to a better performing heterostructure. While this composite proved to be less stable than the bromide counterpart, these results pave the way to further explore this system by improving its stability in water through encapsulations strategies. It is also interesting to confirm the trend as a function of loading, which peaks around 3%, in agreement with the TAS results reported above and with the previous results we provided on the Cs₂Bi₂Br₉/g-C₃N₄ composites.³⁴

Finally, to evaluate the potential of the two best-performing composites for practical photocatalytic applications, 2.5 wt % PhBz₂GeBr₄/g-C₃N₄ and 3 wt % PhBz₂GeI₄/g-C₃N₄ were tested for H₂ evolution from aqueous solutions of glucose and starch, chosen as representative biomass derivatives. Under optimized conditions, H₂ formation was appreciable for both catalysts (Table S1) and was still higher using the iodine-based composite and with a greater H₂ yield in the presence of the monosaccharide, which exhibits faster mass transfer kinetics compared with the branched-skeleton polysaccharide.³⁶ The observed HERs are clearly higher than those of the control samples (no catalysts, <0.008 μmol h⁻¹), demonstrating the effectiveness of such new composites for H₂ photogeneration also from biomasses in solution.

In conclusion, we report on the realization of an intrinsically water stable 2D Ge-based halide perovskite, a material that can sustain suspension in water for several hours. The strategy employed for the realization of such innovative system foresees the use of an extended π-conjugated organic cation (phenylbenzylammonium), which, through intra-layer vDW interactions, sizably stabilizes the resulting bulk perovskite. The material has been tested therefore for simulated solar-light-induced hydrogen evolution from water and aqueous solutions of glucose and starch in combination with a partnering material, which induces the formation of an active heterojunction, leading to very promising HERs up to 1,200 μmol g⁻¹ h⁻¹. This is an outstanding result considering the seminal exploitation of a Ge perovskite for such a purpose and provide a proof of concept for the use of 2D Ge-based MHPs. Most importantly, these findings contribute to expand the rationale behind the intelligent design of intrinsically water-stable MHP phases. The achievement of such an understating would massively impact MHP-based photocatalytic applications but also alternative optoelectronic innovations based on classes of materials such as LED, photovoltaics (PV), and detectors, whose technological declination has been delayed, among other factors, by moisture-exposure weakness of the active material.

**EXPERIMENTAL PROCEDURES**

**Resource availability**

**Lead contact**

Further information and requests for resources and reagents should be directed to and will be fulfilled by the lead contact, Lorenzo Malavasi (lorenzo.malavasi@unipv.it), and the other corresponding authors (andrea.listorti@uniba.it and francesco.ambrosio@unibas.it).
Materials availability
All unique or stable reagents generated in this study are available from the lead contact with a completed materials transfer agreement.

Data and code availability
The authors declare that the data supporting the findings of this study are available within the article and the supplemental information.

Sample preparation
Bulk g-C3N4 has been synthesized from the polymerization of DCD (NH2C(=NH)NHCN, Aldrich, 99%) by the following thermal treatment (under N2 flux): heating (1 °C/min) to 550 °C, isothermal step for 4 h followed by cooling to room temperature (10 °C/min). Synthesis has been carried out in a partially closed alumina crucible. PhBz2GeX4 (X = I and Br) perovskites have been prepared by dissolving GeO2 in HX under stirring and under nitrogen flux. The PhBz2GeX4/g-C3N4 composites have been prepared by adding to the DMF solution containing the perovskites the proper amount of g-C3N4 prepared as described above.

Sample characterization
The crystal structure of the samples has been characterized by room-temperature Cu-radiation XRD acquired with a Bruker D8 diffractometer. Diffuse reflectance spectroscopy (DRS) spectra were acquired in the wavelength range 300–800 nm directly on the powders by using a Jasco V-750 spectrophotometer, equipped with an integrating sphere (Jasco ISV-922). Microstructural characterization of the samples was made using a high-resolution SEM (TESCAN Mira 3) operated at 25 kV. Elemental mapping has been performed on the best-performing and stable composite (cf. section “PhBz2GeBr4 hydrogen generation experiments”), namely 2.5 wt % PhBz2GeBr4/g-C3N4, and the data are reported in Figure S9. Due to the very low amount of metal present in the sample, clear distribution has been obtained only for the Br, C, and N elements, indicating a good distribution of the perovskite in the carbon nitride matrix.

The PL measurements were recorded by means of a Fluorolog-3 spectrofluorometer (HORIBA Jobin-Yvon) equipped with a 450 W xenon lamp as the exciting source and double grating excitation and emission monochromators. All optical measurements were performed at room temperature on powder dispersed samples as obtained from the synthesis without any size sorting treatment. The PL emission spectra were recorded by using an excitation wavelength of 375 nm.

The DT/T was measured on the composites dispersed in Nafion matrix by exciting them with a pulsed laser source (100 fs pulses at 320 nm wavelength) obtained from a kHz regenerative amplifier and an optical parametric oscillator (Coherent Libra and Light Conversion Topas 800). DT/T is then determined with a pump-and-probe technique, where the sample is excited also with fs supercontinuum white pulses, obtained by focusing the fundamental output regenerative amplifier (800 nm) onto a sapphire plate, whose delay with respect to pump pulses is controlled with an adjustable mechanical delay line. DT/T spectra are finally obtained from signal and reference spectra acquired with a couple of CMOS grating spectrometers (Ultrafast Systems Helios).

Hydrogen evolution experiments
H2 evolution experiments were conducted in distilled water containing 10% (v/v) TEAO (Aldrich, ≥ 99%), irradiated in Pyrex glass containers (28 mL capacity, 21 mL
sample). After the addition of the catalyst (1 g L⁻¹), the sample was deoxygenated by Ar bubbling (20 min) to obtain anoxic conditions and irradiated under magnetic stirring for 6 h.

Chloroplatinic acid (H₂PtCl₆, 38% Pt basis), used as precursor for metallic Pt, was from Sigma-Aldrich. Since Pt is in situ photodeposited on the catalyst surface, after Ar bubbling, a small volume from a 15 g L⁻¹ H₂PtCl₆ aqueous solution was added, using a 10–100 µL micropipette, to the catalyst suspension (1 g L⁻¹) directly in the photoreactor. The latter was closed with sleeve stopper septa and was irradiated, as described in the following, achieving simultaneous Pt deposition and H₂ production. Irradiation was performed under simulated solar light (1,500 W Xenon lamp, 300–800 nm) using a Solar Box 1500e (CO.FO.ME.GRA S.r.l., Milan, Italy) set at a power factor 500 W m⁻² and equipped with UV outdoor filter made of infrared (IR)-treated soda-lime glass. Triplicate photoproduction experiments were performed on all samples. The headspace-evolved gas was quantified by gas chromatography coupled with thermal conductivity detection (GC-TCD). The results obtained in terms of H₂ evolution rate are expressed in the paper as μmoles of gas per gram of catalyst per hour (μmoles g⁻¹ h⁻¹). XRD measurements on spent catalysts have been done by filtering the suspensions and recovering the powder, which underwent diffraction measurements.

**Metal leaching tests**

The leaching tests were performed by dispersion of the powders in distilled water under magnetic stirring for 4, 8, 16, and 24 h. Then, the suspension was filtered on 0.2 µm nylon membrane, and the amount of tin in solution was determined by ICP-OES analysis after acidification (1% v/v ultrapure nitric acid).

**SUPPLEMENTAL INFORMATION**

Supplemental information can be found online at https://doi.org/10.1016/j.xcrp.2022.101214.

**ACKNOWLEDGMENTS**

F.A., E.M., and F.D.A. acknowledge support from the Ministero Istruzione dell’Università e della Ricerca (MIUR) and the University of Perugia through the program “Dipartimenti di Eccellenza 2018–2022” (grant AMIS) and from the European Union’s Horizon 2020 research and innovation program under grant agreement no. 764047 of the Espresso project. This work has been financed by the Research Fund for the Italian Electrical System under the contract agreement between RSE S.p.A. and the Ministry of Economic Development - General Directorate for the Electricity Market, Renewable Energy and Energy Efficiency, Nuclear Energy, in compliance with the Decree of April 16th, 2018. A.S. was supported by PON “Ricerca e Innovazione” 2014–2020 FSE, project AIM1809115-2. A.L. acknowledges support from Puglia regional council (grant name: Perseo, CUP: H95F20000890003).

**AUTHOR CONTRIBUTIONS**

L.R., M.M.-L., C.C., and C.T. performed photocatalysis experiments; A.S. and A.P. supervised photocatalysis experiments and analyzed the data; M.M. carried out structural characterization; V.A., S.C., A.M., and A.L. performed part of the optical characterization and XPS experiments; F.A., E.M., D.R., and F.D.A. performed computational modeling work and analyzed the data, R.P., F.P., A.S., and M.S. carried out TAS measurements and related data analysis; and L.M. devised the work.
The manuscript was written by L.M., A.L., and F.A. All authors have approved the final version of the manuscript.

DECLARATION OF INTERESTS
The authors declare no competing interests.

REFERENCES


